

Enhanced Dielectric Performance of HfO₂ Thin Films Via Novel Atomic Layer Deposition Conversion at Production Speed and Efficiency

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Abstract

High quality dielectric ALD films of HfO₂ and AlHfO_x have been deposited at production worthy speed and efficiency. The CRISP HfO₂ process is up to three times faster than other ALD tools and uses up to 29 times less Hf-metal precursor. The dielectric properties of the films are influenced by the type of conversion, pure O₃ or the novel CRISP conversion, and the amount of Al₂O₃ introduced into the bulk HfO₂ film in a nanolaminate scheme. For HfO₂ films deposited at 250°C, Addition of Al₂O₃ at varying amount gave the highest κ , up to 18.9, improved dielectric strength up to 7.8 MV/cm, and reduced leakage current at 60V down to 8.1e⁻¹² J. These results are a critical step to providing improved dielectric ALD films to the compound semiconductor community for dielectric barriers and MIM capacitors.

INTRODUCTION

Hafnium dioxide, HfO₂, is a high- κ dielectric barrier with exceptional stability that has been used in a litany of applications, such as the high- κ metal gate dielectric for transistors.¹ While the dielectric constant is high, ~20-25, the application of HfO₂ can be limited due to high leakage currents or low breakdown voltages. Fortunately, the addition of a secondary dielectric material, such as Al₂O₃ or Ta₂O₅, can improve the deficiencies and open new applications such as MIM capacitors. One technique to provide controlled doping or layering of a secondary oxide is atomic layer deposition, ALD.

ALD is an enabling technology for dielectric barriers over a range of device applications, from gate dielectrics to insulating barriers for a range of compound semiconductor devices such as GaN HEMT and MEMS devices.² Through self-limiting surface chemistry, ALD provides a conformal and pin-hole film of high quality for bulk materials at thin total thickness, <10nm over any device architecture accessible to the process gas. Due to the digital growth nature of ALD, it also has the advantage of creating nanolaminate films with vertically discrete layers of a secondary dielectric to tune the bulk dielectric properties of the full ALD film. This dielectric tuning has been exploited for multiple oxide systems, HfO₂/Al₂O₃,³ Ta₂O₅/ZrO₂ or Al₂O₃,⁴ and others,⁵

with success. HfO₂/Al₂O₃ nanolaminate ALD films have found significant success at MIM-cap devices, both in conventional CMOS Si flow⁶ and compound semiconductor applications, such as GaAs HBT devices.⁷

Two major drawbacks to wider adoption of ALD solutions for MIM-cap applications is the lack of speed and efficiency of the ALD processes. The root cause of this issue is that traditional ALD reactors are stuck in a process chamber pressure-based tradeoff. At high pressure, precursors are used more efficiently, but purge times suffer and at low pressures, purge times improves, but precursor efficiency degrades. Fortunately, Forge Nano has single wafer ALD technology that increases deposition rate and improves precursor efficiency. Enabled by the proprietary reactor design and fast pneumatic dosing valves, the ALD^X tool set enables faster R&D development and higher wafer throughput by breaking the purge-precursor efficiency tradeoff.⁸ With deposition rates up to 12 nm/minute and precursor efficiencies up to 85%, application of high-quality ALD dielectric barriers for compound semiconductors devices is more available to the market.

RESULTS AND DISCUSSION

To improve the properties of ALD HfO₂ films, a comparison of a standard ALD conversion step, using O₃ alone, with a novel conversion step, referred to as a CRISP conversion, was performed on the ALD^X production tool, Apollo. The CRISP conversion entrains a small amount of non-metal catalyst into the conversion precursor flow and provides a new reaction pathway for ALD surface chemistry to proceed. The ability to control the addition of small amounts of catalyst in the conversion step, not quench the O₃, is a unique feature of the ALD^X toolset.

As shown in Table I, for ALD processes at 250°C, the CRISP process increased GPC by 29%, reduced Hf-metal precursor consumption by over three times, over the O₃ only conversion. CRISP directly improves the wafer throughput and precursor consumption. The 18-minute deposition time for 20 nm of HfO₂ is the fastest the authors have observed in literature. Critically, the film properties also improved, as shown by a density increase of 7%, more ideal O:Hf stoichiometry, and a decrease of 44% in C impurities. Both

films are crystalline as deposited at 250°C and a simple Scherrer analysis shows that the CRISP process has larger crystal grains and a higher proportion of crystalline to amorphous film. Taken together, these data suggest that the overall quality of the HfO₂ film is superior to a conversion with O₃ alone.

TABLE I
HfO₂ FILM PROPERTIES FOR O₃ AND CRISP CONVERSION ON THE ALD^x TOOLSET DEPOSITED AT 250°C

	O ₃ Only	CRISP	Difference
GPC (Å/cy)	0.77	0.99	29% faster
20 nm Film Deposition (hours)	0.98	0.3	3.3x faster
Hf Consumption/ 20 nm (Torr·s)	56	16	3.4x less
Density (g/cm ³)	8.3	8.9	7% more dense
O:Hf Ratio (XPS)	2.4	2.2	0.2 more ideal
C content (%)	3.2	1.8	1.4% less

Since the HfO₂ CRISP process takes advantage of surface catalysis, which provides access to different surface reaction mechanisms, the operating temperature for ALD is wider, as shown in Fig. 1 below. A stable ALD process is accessible for the HfO₂ CRISP process from 100 to 300°C, providing compatibility with a range of materials and device thermal budgets. The increase in GPC, coupled with the decrease in RI at 633 nm as the deposition temperature decreases, suggests that a combination of lower film density and higher impurity incorporation occurs in film with lower temperatures.

Using a proprietary reactor design and ALD dose valves, the ALD^x toolset allows for extremely fast cycle time and drastic reduction in precursor consumption. For HfO₂ CRISP, these tools have higher deposition rate than both batch ALD and PEALD tools by 70⁹ to 300%¹⁰ respectively. This faster deposition rate correlates to a higher wafer throughput of about 2-3 times. Notably, this unique HfO₂ process on the ALD^x toolset reduces Hf precursor consumption by 6⁹-29¹⁰ times, when compared to published data for batch and PEALD tools. Precursor consumption is often the highest input for ALD tool OpEx, so the HfO₂ CRISP process provides a significant reduction in OpEx.

With this promising HfO₂ ALD film as a foundation, introduction of a small amount of Al₂O₃ periodically into the growing film was explored. Tuning the amount of the Al₂O₃ secondary oxide in the HfO₂ bulk has been shown in previous internal data to enhancement dielectric strength and reduce leakage current density, without sacrificing κ for bulk HfO₂ film. Full dielectric measurements were undertaken and compared with the standard O₃ conversion. A simple MIM setup was constructed of 100 nm of Au film on 4" Si wafers,

200nm dielectric (HfO₂ or AlHfO_x) and 100 nm of Au deposited as 2- or 3-mm electrodes patterned with a shadow mask by magnetron sputtering. Both Au films had a thin (~10 nm) layer of Ti as an adhesion layer prior to sputtering. Electrode size and regularity were checked on all samples by optical microscopy. Measurements were performed at room temperature on a Trek 601C for dielectric breakdown, an Agilent E4908A Precision LCR Meter for κ, and a Keithley 6517A electrometer for leakage current at 60V.

The results for these MIM-based measurements of the bulks HfO₂ ALD films can be found in Table II below. Interestingly, the higher density and lower impurity content observed in the CRISP HfO₂ does not translate to higher κ or dielectric strength, but does improve leakage current, for the bulk film over the O₃ process. This is unexpected as normally density and composition correlates directly with dielectric characteristics. The authors suspect that the high thickness, 200 nm, of the film deposited provided more grain boundaries to form, artificially degrading the performance of the CRISP film.

Fabrication of nanolaminate films, the addition of 1-10% of Al₂O₃ layered periodically into the growing HfO₂ film, improved the dielectric properties, depending on the thickness of the Al₂O₃ layer. Specifically, the addition of just 1% Al₂O₃ increased κ to 19.2 for the CRISP process, giving the highest value for the Al₂O₃ series with both CRISP and O₃ conversions. This data further supports a κ measurement artifact for both bulk HfO₂ films, as the κ of the bulk HfO₂ O₃ and CRISP films are artificially low, 17.6 and 16.9, and the addition of Al₂O₃, which has a κ around 8 at these deposition temperatures, increase the κ. It is more common that the addition of lower κ Al₂O₃ should decrease κ from the bulk HfO₂ value or leave it unchanged, in the absence of other factors. The authors speculate that the small amount of Al₂O₃ disrupts the crystallinity of the growing HfO₂ film, allowing for the improvement in κ, dielectric strength, and leakage current density.

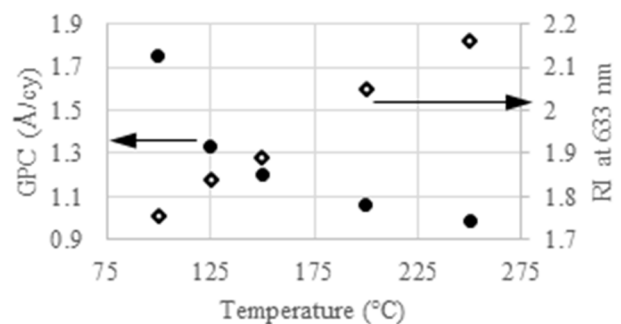


Fig. 1. GPC (solid circles, left axis) and RI (empty diamonds, right axis) for the HfO₂ CRISP process from 100 to 250°C.

All the HfO₂ films had impressive dielectric strength (dielectric field breakdown), with the highest being the 5%

Al₂O₃ in the HfO₂ O₃ film at 7.8 MV/cm. Dielectric strength appeared to be less correlated with Al₂O₃ content and more correlated with conversion process. Increasing the Al₂O₃ content in the CRISP series had no correlation with dielectric strength and both O₃ conversion samples were higher than CRISP conversion samples, regardless of Al₂O₃ content.

TABLE II

HfO₂ film Dielectric properties for O₃ and CRISP conversion on the ALD^x toolset

Film	κ (At 10 ⁶ Hz)	Dielectric Strength (MV/cm)	Leakage Current Density at 60V (J – A/cm ²)
HfO ₂ – O ₃	17.6	7.7	3.3e ⁻⁹
HfO ₂ – CRISP	16.2	6.9	2.3e ⁻¹⁰
1% Al ₂ O ₃ 99% HfO ₂ – CRISP	19.2	7.5	3.3e ⁻¹¹
5% Al ₂ O ₃ 95% HfO ₂ – CRISP	16.9	7.0	1.1e ⁻¹¹
5% Al ₂ O ₃ 95% HfO ₂ – O ₃	17.0	7.8	3.9e ⁻¹¹
10% Al ₂ O ₃ 90% HfO ₂ – CRISP	18.9	7.4	8.1e⁻¹²

Showing the flexibility of the nanolaminate ALD technique, one can minimize leakage current density by adding 10% Al₂O₃ to the HfO₂ CRISP film. This improves the leakage by 1.5 orders of magnitude, to 8.1e⁻¹² J at 60V, from the bulk film, 2.3e⁻¹⁰ J, while only minimally impacting the κ , 18.9, or dielectric strength, 7.4 MV/cm. Addition of even 1% of Al₂O₃ into the bulk HfO₂ CRISP film improved leakage current by almost an order of magnitude, from 2.3e⁻¹⁰ to 3.3e⁻¹¹ J. A similar effect is observed when 5% Al₂O₃ is added to the HfO₂ O₃ film. When a comparison is made between a 5% Al₂O₃ incorporation level of the O₃ and CRISP conversion, both films appear well suited for MIM cap applications. Both have a high dielectric constant, ~17, the O₃ conversion has a slightly higher dielectric constant and dielectric strength, but the CRISP conversion has a lower leakage current density.

Further tests can be undertaken to explore how often Al₂O₃ film is incorporated into the growing HfO₂ film by vertically adjusting the incorporation. In these tests, a static amount of HfO₂ film was deposited at each Al₂O₃ incorporation amount, enforcing a thicker Al₂O₃ layer at the same periodicity. Alternatively, one can keep the Al₂O₃ film thickness static and incorporate it more frequently into the growing HfO₂ film to increase incorporation content. It is expected that the method of incorporation will also impact dielectric properties of the film, providing further control over the dielectric properties.

Finally, we expect increasing the deposition temperature of the films to improve dielectric properties across the board. As such, increasing the temperature of deposition up to 300°C or adding a post deposition anneal will also be explored in future work.

CONCLUSIONS

Overall, our novel HfO₂ ALD O₃-based processes have higher throughput and lower precursor consumption than comparable ALD techniques, batch ALD or PEALD. The film properties can be tuned with the addition of a CRISP component to increase deposition speed and film density, while reducing impurity incorporation and Hf precursor consumption, while opening a wider temperature range for sensitive applications. At 250°C the films high quality dielectric films are deposited and using the ability of ALD to form nanolaminate structures, the dielectric properties of the HfO₂ film can improved and tuned. With varying amounts of Al₂O₃, κ can be increased from 16.2 to 19.2, the dielectric strength can be increased from 6.9 to 7.8 MV/cm, and the leakage current density can be reduced from 3.3e⁻⁹ to 8.1e⁻¹² J at 60V. In the future, full characterization in GaN HEMT devices is planned for both the HfO₂ – O₃ and HfO₂ – CRISP processes.

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ACRONYMS

HfO₂: Hafnium Dioxide
κ: Dielectric constant
Al₂O₃: Aluminum Dioxide
Ta₂O₅: Tantalum Pentoxide
ZrO₂: Zirconium Oxide
ALD: Atomic Layer Deposition
MIM-cap: Metal-Insulator-Metal Capacitor
GaN: Gallium Nitride
HEMT: High-Electron-Mobility Transistor
HBT: Heterojunction bipolar device
MEMS: Microelectromechanical Systems
CMOS: Complementary metal-oxide semiconductor
GaAs: Gallium Arsenide
CRISP: Catalyzing Reactions for Induced Surface Process
OpEx: Operation Expense
O₃: Ozone
PEALD: Plasma Enhanced Atomic Layer Deposition
GPC: Growth per Cycle
RI: Refractive Index
LCR: Inductance (L), Capacitance (C), and Resistance (R) meter